Characterizing Phase Transitions in the Perovskites PbTiO$_3$ and BaTiO$_3$
Using Perturbed-Angular-Correlation Spectroscopy*

Gary L. Catchen, Edward F. Hollinger**, and Todd M. Rearick***
Department of Nuclear Engineering and Electronic Materials and Processing Research Laboratory,
The Pennsylvania State University University Park, Pennsylvania 16802 USA

Z. Naturforsch. 51 a, 411–421 (1996); received November 20, 1995

Perturbed-angular-correlation (PAC) spectroscopy was used to measure in ceramic samples of PbTiO$_3$ and BaTiO$_3$, the temperature dependence of the Ti-site electric-field gradients (EFGs) at temperatures very close to the ferroelectric-to-paraelectric transition temperatures $T_c$. The samples were doped with small amounts of Hf that carried the $^{181}$Hf $\rightarrow$ $^{181}$Ta probe radioactivity. A high-frequency nuclear quadrupole interaction that decreases very little as the temperature approaches $T_c$, characterizes the PbTiO$_3$ transition. The tetragonal and cubic phases for PbTiO$_3$ appear to coexist over a temperature interval of $8 \pm 1$ K, and the transition shows a thermal hysteresis of about 4 K. In contrast, a lower-frequency interaction that decreases rapidly as temperature approaches $T_c$, characterizes the BaTiO$_3$ transition. Both phases of BaTiO$_3$ appear to coexist over an interval of about 2 K, and the thermal hysteresis is about 1 K. At temperatures above $T_c$, both PbTiO$_3$ and BaTiO$_3$ show weak, non-vanishing Ti-site EFGs. Although, for BaTiO$_3$, this effect limits the accuracy with which critical effects can be measured, we estimate a power-law exponent $\beta = 0.21 \pm 0.05$, which most likely is somewhat lower in magnitude than the actual critical exponent. For the explanation of our observations we assume the existence of a distribution of $T_c$-values. This distribution would arise because the crystals could have spatially non-uniform distributions of nucleation sites, which for PbTiO$_3$ and BaTiO$_3$ could be point defects.

Key words: Ferroelectric, perovskite, nuclear electric-quadrupole, perturbed-angular-correlations, phase transition.

I. Introduction

During the past decades electric- and magnetically-ordering phase transitions, that occur in metal-oxide perovskite and related crystals, have been investigated. To characterize these transitions, macroscopic quantities such as heat capacities, electric and magnetic susceptibilities [1], and neutron[2] and x-ray [3] diffraction-derived parameters have been measured. However, by their very nature these quantities represent gross averages over the atomic-scale properties of these crystals. To measure the atomic-scale properties directly, spectroscopy has been applied.

To investigate ferroelectric transitions, Raman scattering has been used [4]. More recently, impulsive-stimulated Raman scattering [5] has been developed to provide more information about soft and heavily-damped vibrational modes that conventional frequency-resolved Raman scattering is not very sensitive to. But, because crystal vibrations represent collective motions that extend over the entire crystal, these light-scattering techniques do not provide detailed information about the effects of local fields that occur at specific lattice sites in a crystal. However, in electrically-ordered crystals, the effects of local crystal fields can provide important, unique information that can be used to distinguish, for example, between displacive and order-disorder mechanisms of ferroelectric transitions [6]. In magnetically-ordered crystals, the crystal-field information can provide, e. g., detailed information about antiferromagnetic transitions [7].

To investigate the effects of crystal fields, spectroscopic techniques that measure hyperfine inter-
actions have been used, mostly perturbed-angular-correlation (PAC) [8] and Mössbauer-effect (ME) spectroscopies. Primarily magnetically-ordered perovskites have been investigated using $^{57}$Fe [9] and $^{99}$Ru [10] ME spectroscopy, but several non-magnetic, ferroelectric perovskites have also been investigated [11]. Studies of magnetic crystals have focused on characterizing exchange interactions [10] and measuring critical phenomena [9].

The PAC technique involves substituting a small number of radioactive probe ions into a specific site in the crystal of interest and measuring the hyperfine interaction at that site [8]. In non-magnetic crystals one measures the nuclear electric-quadrupole interaction and in cubic magnetic crystals the magnetic-dipole interaction. In lower-symmetry magnetic crystals one measures the combined nuclear magnetic-dipole and nuclear electric-quadrupole interaction [12]. During the measurement, the radioactive probe undergoes beta decay and populates an excited level in the daughter nucleus. This excited level decays by successively emitting two $\gamma$-rays. The first $\gamma$-ray populates an intermediate quantum level that has a long lifetime, and the second $\gamma$-ray depopulates this level. During the lifetime of the intermediate quantum level, the nuclear quadrupole moment (in nonmagnetic crystals) interacts with the electric-field gradient (EFG) at the probe nucleus. This interaction causes the nucleus to reorient and to emit the second $\gamma$-ray into a different region of space than where it would have emitted the second $\gamma$-ray if no interaction had occurred. This measurable effect is the basis for PAC spectroscopy. Specifically, two tractable probes are $^{181}$Hf $\rightarrow$ $^{181}$Ta and $^{111}$In $\rightarrow$ $^{111}$Cd, which both have spin I = 5/2 intermediate levels and which have intermediate-level half lives of 10.8 nsec and 85 nsec, respectively. To investigate Ti-containing perovskites such as PbTiO$_3$ and BaTiO$_3$, the $^{181}$Hf $\rightarrow$ $^{181}$Ta probe has the appropriate chemistry to substitute into the Ti sites. However, for this probe the $^{181}$Ta nucleus undergoes the actual hyperfine interaction with the extranuclear environment.

Measurements of static probe-site EFGs yield sharp spectral lines when each probe nucleus in the ensemble interacts with an identical EFG. This situation occurs when the probes occupy a specific site in a nearly perfect crystal. The measurements yield broadened lines when point defects lie close to the probe site and each probe in the ensemble interacts with a somewhat different EFG. When this situation occurs, a distribution of EFGs characterizes the interaction. However, even an infinitely-broad distribution of static EFGs does not completely destroy the angular correlation. In contrast to static EFGs, the measurement of EFGs that fluctuate on the PAC experimental time scale, which, for the $^{181}$Hf $\rightarrow$ $^{181}$Ta probe, varies from $\approx$ 0.1 to 80 nsec, yields spectral lines that show increasing attenuation as the intermediate-level lifetime increases. An interaction with a fluctuating EFG eventually causes the destruction of the angular correlation because the fluctuations change the direction of the quantization axis; we refer to this process as “nuclear-spin relaxation”.

PAC spectroscopy has been used to investigate both magnetically-ordered and electrically-ordered perovskites. The investigations of magnetic crystals using non-magnetic probes primarily have involved measuring supertransferred hyperfine fields at the nuclear probe sites via the magnetic-dipole interaction [13, 14]. These fields arise when electron-spin density is transferred through ligand p orbitals to 4s and 5s orbitals of the probe ion, which is a non-magnetic impurity ion that occupies a metal-ion crystal site. These fields provide unique information about the spatial extents of atomic wave functions in the crystal [14]. The investigation of ferroelectric and antiferroelectric crystals has involved measuring EFGs at probe sites via the nuclear quadrupole interaction (NQI) [15].

Before the mid-1980’s, a variety of ferroelectric and antiferroelectric titanates, zirconates, and hafnates had been investigated using PAC spectroscopy, which had involved using primarily the $^{181}$Hf $\rightarrow$ $^{181}$Ta probe, although the $^{44}$Ti $\rightarrow$ $^{44}$Sc and $^{111}$In $\rightarrow$ $^{111}$Cd probes had also been used [15]. From these investigations, the relationships had been developed between the crystal structure and the parameters that characterize the measured hyperfine interaction, which occurs at a specific crystal site. During this era, Yeshurun, Schlesinger, and Havlin [16] performed the most detailed investigation on PbHfO$_3$, in which they measured and analyzed critical phenomena associated with two antiferroelectric transitions. This work established that PAC spectroscopy can be used to characterize critical phenomena associated with ferroelectric and antiferroelectric transitions.

To use PAC spectroscopy to measure critical effects that are associated with ferroelectric and antiferroelectric transitions has been a difficult undertaking. In contrast to measuring critical effects that accompany magnetic transitions, the effects, as they are manifest in the PAC measurements, are difficult
G. L. Catchen et al. • Characterizing Phase Transitions in the Perovskites PbTiO₃ and BaTiO₃

II. Experimental Details

Samples of PbTiO₃ and BaTiO₃ were prepared using a resin-intermediate process. To a solution of ethylene glycol and citric acid, the titanium precursor Ti(OC₃H₇)₄ and either Ba(OOCH₃)₂ or Pb(OOCH₃)₂ were added. The ¹⁸¹Hf activity was carried by a dilute aqueous solution of HIOCl₂, and it was added to the PbTiO₃ preparation at ≈ 0.25 at.% of the Ti concentration and to the BaTiO₃ preparation at ≈ 0.08 at.% of the Ti concentration. The resin was formed by heating the solution that contained the chelated metal ions. The resin was subsequently pyrolyzed and then calcined at 1070 K. As indicated by x-ray powder diffraction, the PbTiO₃ compound formed during the calcination, and it received no further processing. The calcined powder for the BaTiO₃...
preparation was pressed into small pellets and sintered in air at \( \approx 1800 \) K for about an hour. The x-ray powder diffraction patterns were measured on small amounts of powder taken from the radioactive samples on which the PAC measurements were made, and the samples were found to be phase pure to within several percent.

Recent reviews present most of the experimental details involved with performing and analyzing the PAC experiments [8, 22]. The experimental time distributions were measured using a four-BaF\(_2\)-detector PAC spectrometer which has a time resolution of \( \approx 800 \) psec full width at half maximum. A specially-designed furnace and temperature control system was used to maintain the samples at constant temperature, \( \pm 0.03 \) K, during the measurement periods of 0.5 - 1 day. Also the controller was adjusted to avoid significant overshoots and undershoots when the sample temperature was either increased or decreased. We note that the BaTiO\(_3\) measurements were performed using the first-design furnace that was large in diameter. The source-to-detector distances were \( \approx 9 \) cm, which required using relatively strong sources. The resulting measurements included prompt coincidences that destroyed the first 1 - 2 nsec regions of the time distributions. The PbTiO\(_3\) measurements were performed using a second-generation furnace that is smaller in diameter. The corresponding source-to-detector distances are \( \approx 5 \) cm, which reduces the effects of prompt coincidences. The perturbation functions \( A_{22}G_{22}(t) \) were determined from the time distributions. A two-site model for static nuclear-electric-quadrupole interactions was sufficient to analyze the measured perturbation functions:

\[
-A_{22}G_{22}(t) = \sum_{j=1}^{2} A_j \left[ \sum_{k=1}^{3} S_k \exp(-\delta_j\omega_k t) \right] + A_3.
\]

III. Results and Discussion

Figure 1 presents several perturbation functions for PbTiO\(_3\) that were measured at temperatures very close to \( T_c \) and that illustrate the effects of the tetragonal-to-cubic transition on the Ti-site EFG. The perturbation functions measured above \( T_c \approx 760.5 \pm 0.5 \) K show little structure and represent very weak perturbations. Perturbation functions measured at temperatures several hundred degrees higher show the same shape. Thus, above \( T_c \), the Ti-site EFG becomes temperature independent. To fit the above \( -T_c \) perturbation functions, we used (1) and constrained the frequencies so that \( \eta \approx 0 \). Typically, the fits required using admixtures in the fitting function of a very-low-frequency, highly-damped static-interaction term and a zero-frequency term \( (A_1 \text{ in (1))}. These conditions strongly suggest that point defects, which give rise to weak static interactions, occupy positions near some of the probes and the remainder of probes occupy defect-free environments, which give rise to no interactions.

The perturbation functions measured below \( T_c \) show the features of at least two interactions, a relatively strong well-defined interaction that increases in strength only slightly as temperature decreases from \( T_c \), and a very weak interaction. To fit the below \( -T_c \) perturbation functions, we used (1) and constrained the frequencies so that \( \eta = 0 \). Typically, the fits required using, in the fitting function, two static-interaction terms and a zero-frequency term. The term that represented the strong well-defined interaction yielded unique parameters that vary systematically with temperature changes and gave \( V_{\perp,\parallel} \approx 4 \times 10^{17} \) Vcm\(^{-2}\) and \( \delta \approx 0.15 \pm 0.05 \). The term that represented the very weak interaction along with the zero-frequency term yielded parameters that were...
Fig. 1. Perturbation functions for a PbTiO₃ powder sample at increasing temperatures. The solid lines represent fits of (1) to the data points. For this set, $T_c = 760.5 \pm 0.5$ K. The two functions measured at 761.5 K and 765.0 K represent the paraelectric phase, which has nominal cubic symmetry but nonvanishing EFGs.

difficult to characterize. At temperatures very close to $T_c$ (that correspond, for example, to some of the perturbation functions shown on the right side of Figure 1), this component gave $V_{zz} \approx 0.1 \times 10^{17}$ Vcm⁻² and $\delta \gg 1$; and, at temperatures further away from $T_c$ (that correspond to the perturbation functions shown on the left side of Fig. 1), this component gave $V_{zz} \approx 1 \times 10^{17}$ Vcm⁻² and $\delta < 1$. Generally, acceptable fits
Fig. 2. Parameter summary for the analysis of the measurements on a PbTiO$_3$ powder sample. The $V_{zz}$-values correspond to the high-frequency interaction that occurs below $T_c$, and the $f$-values represent the corresponding fractions of probes that underwent this interaction. The circles represent fits that consist of a single static-interaction term plus a zero-frequency term in (1). The squares ($T$ increasing) and the triangles ($T$ decreasing) represent fits that consist of two static-interaction terms plus a zero-frequency term. The $V_{zz}$-values could not be obtained with $A_1 = 0$. Thus, we associate the strong well-defined interaction with the tetragonal, ferroelectric phase, because the corresponding EFG parameters resemble those that characterize the Ti-site interaction at temperatures well below $T_c$, in which $V_{zz}$ is large, $\eta$ is near zero, and $\delta$ is moderate (at laboratory temperature, $V_{zz} = 14.5 \pm 0.1 \times 10^{17}$ Vcm$^{-2}$, $\eta = 0.02 \pm 0.02$, and $\delta = 0.04 \pm 0.01$). The other interaction cannot be accurately characterized over the $\approx 8$ K range where multiple Ti-site interactions take place, and it does not completely resemble the interaction observed above $T_c$. Therefore, we speculate that complex physical changes could occur in sample regions where the probes appear to undergo this poorly-defined interaction; and the static model (1), which includes no spin-relaxation effects, does not provide a good representation of these effects.

Figure 2 presents a summary of the $V_{zz}$ and site-occupancy fraction $f$ values that represent the aforementioned strong, well-defined interaction. The values of the EFG component $V_{zz}$ change by less than 20% over the indicated 15 K interval, and these values do not depend on whether the measurements involved either successive increases or successive decreases of the temperature. The site-occupancy values show clear evidence for the onset of the tetragonal-to-cubic transition. Therefore, we associate the measurement of these two interactions with the coexistence of the tetragonal and cubic phases within the sample. The $f$-values change from one to zero over an $8 \pm 1$ K range. However, these $f$-values can only be accurately

Fig. 3. Perturbation functions at various temperatures for a sintered-ceramic sample of BaTiO$_3$. Solid lines: fits of (1) to the data points. For this set, $T_c = 390.3 \pm 0.4$ K. The function at 389.2 K is typical for the paraelectric phase, in which the EFGs do not vanish.
measured over the range \(0.2 \leq f \leq 0.8\). The locus of \(f\)-values that corresponds to measurements performed at successively higher temperatures is displaced by \(5 \pm 1\) K from the locus that corresponds to measurements performed at successively lower temperatures. This difference indicates that considerable thermal hysteresis accompanies this transition. The error bars indicated on the \(f\)-values appear to be very large. These error bars are determined using standard non-linear regression analysis. As such, they represent confidence intervals that reflect the uniqueness of the derived parameters. Because non-unique parameters describe one of the two interactions that \(f\)-values represent, these error bars tend to appear to be large. Nonetheless the \(f\)-values are not indeterminate. They vary systematically and appear to be self-consistent.

Figure 4 summarizes the temperature dependences of \(V_{zz}\) and \(f\)-values. For the \(V_{zz}\) and \(f\)-values measured at increasing temperatures, \(T_c = 390.3 \pm 0.4\) K, and for those measured at decreasing temperatures, \(T_c = 389.2 \pm 0.4\) K, the fits represent the sum of a static-interaction term and a zero-frequency term.

Discussion

A large amount of experimental information [3, 5, 21] indicates that the tetragonal-to-cubic transition in BaTiO\(_3\) has order-disorder features. If this transition were strictly first-order in character, we would not expect to observe evidence for critical effects. Because the transition does show order-disorder features that can be associated with continuous transitions, we, therefore, might expect the transition to show evidence for critical effects. The power-law dependence of a quantity proportional to the order parameter for the transition would constitute this type of evidence, and the corresponding exponent would represent the critical exponent \(\beta\) [23]. For the ferroelectric-to-paraelectric transition, the electrical polarization of the crystal represents the order parameter [1, 23].

We posit the hypothesis that the EFG component \(V_{zz}\) is proportional to the order parameter, which is the
To further analyze the site-occupancy information, we form the ratio, \( \frac{1}{1 - f} \), that represents the ratio of electrical polarization of the crystal. To test the hypothesis, first we determine the \( T_c \) values from the \( V_{zz} \) temperature dependences; one \( T_c \)-value corresponds to the temperature-increasing series of measurements and one corresponds to the temperature-decreasing series. Second, we calculate the reduced temperature values \( T_r = 1 - \left( \frac{T}{T_c} \right) \) for each set of measurements and plot both sets of \( (T_r, V_{zz}) \) data points on logarithmic coordinates. Figure 5 represents this analysis, and the derived power-law exponent is \( 0.21 \pm 0.05 \).

We determined the apparent uncertainty in this value by using limiting slopes. In addition, the choice of \( T_c \) values produces a non-explicit uncertainty that has approximately the same magnitude. Despite these sizable uncertainties, the dependence of \( \log V_{zz} \) on \( \log T_r \) shows a strong linear correlation, which we also observed in several earlier sets of measurements [24, 25]. The derived \( \beta \)-value is lower than the values that we would expect to observe during a transition in a three-dimensional crystal, \( 0.3 < \beta \leq 0.4 \) [23]. However, the \( V_{zz} \) values measured above \( T_c \) represent, we believe, the EFGs generated by static point defects; and, therefore, the \( V_{zz} \) values just below \( T_c \) represent both the EFGs generated by the ferroelectric lattice distortion and the EFGs generated by the point defects. At these temperatures, the defect contribution to the total values is significant. Therefore, a correction should be made to the below \( -T_c \) \( V_{zz} \) values for the EFGs generated by the defects. Calculating the average above \( -T_c \) \( V_{zz} \) value and subtracting it from each of the below \( -T_c \) \( V_{zz} \) values would produce too large of a correction, and \( V_{zz} \) would decrease too rapidly, i.e., \( \beta \) would become too large. Performing no correction causes \( V_{zz} \) to decrease too slowly as \( T_r \) decreases, and \( \beta \) would become too small. We know from previous experience [24, 25] that subtracting a \( V_{zz} \)-value of one-half of the average above \( -T_c \) \( V_{zz} \) value would give \( \beta \approx 0.3 \), which would be a reasonable value. Therefore, the power-law dependence of \( V_{zz} \) on \( T_r \) shows the qualitative features that we expect critical effects in BaTiO₃ to exhibit. But we need to develop a model that would provide a means to separate the contribution of defect-generated EFGs to the below-near-\( T_c \) \( V_{zz} \)-values.

Because the tetragonal-to-cubic transitions for both PbTiO₃ and BaTiO₃ have been identified traditionally as being first-order transitions [1], we must consider the physical significance of the corresponding site-occupancy-fraction temperature dependences. If these transitions were truly first-order, then we would expect the two phases, which are the tetragonal and cubic phases in this case, to coexist only at a specific temperature \( T_c \). We examine these temperature dependences, which Figs. 2 and 4 show respectively for PbTiO₃ and BaTiO₃, and we observe that the two phases coexist over relatively broad intervals of \( \approx 8 \) K and \( \approx 2 \) K, respectively. This observation is not consistent with the character of first-order transitions. However, these transitions show thermal hystereses of \( \approx 4 \) K and \( \approx 1 \) K, respectively, which do characterize first-order transitions.

To further analyze the site-occupancy information, we form the ratio, \( f/(1 - f) \), that represents the ratio...
of the probe concentration in the tetragonal phase to the probe concentration in the cubic phase (more accurately, the probe concentration in all other environments). This ratio shows a reasonably strong linear correlation with inverse temperature. Figures 5 and 6 show this dependence for PbTiO$_3$ and BaTiO$_3$, respectively. We observe that the slopes of the lines are rather large, since the inverse temperature scale spans only several degrees. The slopes that correspond to the values measured at successively higher temperatures are approximately equal to the slopes that correspond to the values measured at successively lower temperatures. Although graphs of this type are commonly referred to as “Arrhenius plots”, several physical interpretations of the slope of the line may be invoked. One interpretation involves the kinetics of a process. When the logarithm of a quantity proportional to the rate of a process is plotted as a function of inverse temperature, the analysis of the slope gives the activation energy $\Delta E$, which represents the height of a kinetic barrier. A very different interpretation arises when the process is at chemical equilibrium. Under this condition, when the logarithm of the equilibrium constant $K$ is plotted as a function of inverse temperature, the analysis of the slope gives the standard-state Gibbs free-energy change for the process $\Delta G^0$, which represents the free-energy required to convert a specific quantity of reactants into products.

For PbTiO$_3$ and BaTiO$_3$, the average slopes of the lines shown in Figs. 5 and 6 give energies of 21 ± 5 eV and 11 ± 3 eV, respectively. These energies are much too large to represent either activation energies or Gibbs free-energies. Therefore, the Arrhenius-type plots, which Figs. 6 and 7 show, represent a convenient way to linearize the probe-concentration ratios that characterize these transitions, but they do not yield energies that describe any known physical process in crystals. To analyze the probe-concentration ratios, which represent the coexistence of both phases at the corresponding temperatures, we attribute the origin of this coexistence of two phases to the effects of a distribution of $T_c$-values within the sample. According to this hypothesis, the actual transition temperature would vary throughout the crystal. At any specific temperature within a temperature interval that is very close to $T_c$, the temperature in some regions of the crystal would exceed $T_c$; and, in other regions, the temperature would not exceed $T_c$. Thus, we would observe a different probe-concentration ratio at each temperature within the interval. Our colleagues at Brigham Young University [26] have proposed this hypothesis, and they have developed a specific model that gives the transition temperatures a Gaussian distribution, which has a mean transition temperature $T_c$ and a standard deviation $\sigma$. Using this model, the fraction of probes that undergo the above $-T_c$ interaction at temperature $T$ is given by the fraction of transition temperatures $F(T)$ that occur in the temperature interval: $-\infty < T' < T$, in which $T'$ is the random variable, namely the local transition temperature. The cumulative distribution function gives the site-occupancy fraction

$$f = 1 - F(T) = \frac{1}{2} \left[ 1 + \text{erf} \left( \frac{(T - T_c)}{\sqrt{2} \sigma} \right) \right]. \quad (2)$$

If we consider values of $T$ that are close to $T_c$, which is a physically realistic situation, then the expression for the probe-concentration ratio becomes

$$\ln(f/(1-f)) = -\frac{2\sqrt{2}}{\sqrt{\pi} \sigma} T_c + \frac{2\sqrt{2}k}{\sqrt{\pi} \sigma kT_c} T_c^2, \quad (3)$$

in which $k$ represents the Boltzmann constant. We recognize that the apparent energies, either $\Delta E$ or $\Delta G^0$,
when divided by $k$, are equal to $2\sqrt{2T_c^2/\pi\sigma}$; and the corresponding standard deviations are equal to $3 \pm 1$ K for PbTiO$_3$ and $2 \pm 1$ K for BaTiO$_3$.

This model gives estimates of the variation in the transition temperature throughout the sample. The more important question is what effect gives rise to the variation in the local $T_c$-values, which we appear to observe. We offer the following explanation. The transition takes place when nucleation and growth occur at nucleation sites. If the nucleation sites were in fact uniformly distributed throughout the sample, then we would expect to observe no variation in the transition temperature. If, however, the nucleation sites were actually species that would be present in low concentrations such as point defects, then the local concentrations of these nucleation sites would be inhomogeneous. Because the nucleation sites would be present in relatively small numbers, their corresponding spatial (not temporal) distributions would be subject to statistical fluctuations. Therefore, the local transition temperatures would depend on the local nucleation-site concentration. Variations in the local transition temperature would give rise to the apparent coexistence of both phases that we observe in the sample. In addition to the site-occupancy measurements, which provide direct evidence for the apparent coexistence phenomenon, the above $-T_c$ perturbation functions for both PbTiO$_3$ and BaTiO$_3$ show related evidence. Specifically, these perturbation functions represent two types of probe environments, because accurate fits could only be obtained when a zero-frequency term is included with a static-interaction term in the fitting function. Therefore, above $-T_c$, some probes undergo very weak interactions that nearby defects produce, and some probes undergo no interactions, which indicate the absence of nearby defects. This observation indicates that the point-defect concentrations are not uniform. If these defects were the nucleation sites, then they would give rise to local differences in the transition temperature, and the defects would be responsible for major aspects of the character of these transitions.

IV. Conclusions

We have measured the temperature dependences of the Ti-site EFGs in samples of PbTiO$_3$ and BaTiO$_3$ during the tetragonal-to-cubic transitions. For both compounds, the high-temperature paraelectric phases show very-weak, non-vanishing, temperature-independent Ti-site EFGs. At temperatures very near but below $T_c$, we can separate the EFG produced by the tetragonal crystal distortion associated with the ferroelectric phase from EFGs associated with the paraelectric phase. For BaTiO$_3$, at temperatures very close to $T_c$, we cannot completely separate the effects produced by the ferroelectric distortion from those associated with the paraelectric phase. This limitation prevents us from deriving accurate values of the critical exponent $\beta$ from the power-law dependence of $V_z$ on $T$. Despite this limitation, the linear correlation of $\log V_z$ with $T$ may be evidence for the occurrence of critical fluctuations during this nominally-first-order transition. The transitions for both PbTiO$_3$ and BaTiO$_3$ show some thermal hysteresis, which is expected to occur during these first-order transitions, but both compounds also show the coexistence of both phases over a short but significant temperature interval, which is not expected to occur during strictly first-order transitions. Associated with this coexistence phenomenon is an apparent linear dependence of the logarithm of the site-concentration ratio $f/(1-f)$ on inverse temperature. We are able to explain this strong correlation using a model that associates a distribution of transition temperatures $T_c$ with each transition. We attribute the presence of the $T_c$ distributions to inhomogeneous distributions of nucleation sites in the samples, which could be point defects. These results indicate that further experimental and theoretical investigation of phase transitions in perovskite crystals should be carried out.

Acknowledgements

We greatly appreciate the insight into these complex measurements and the wisdom that Professor William E. Evenson and Professor David Allred of Brigham Young University have provided. Their timely development of the $T_c$-distribution model enabled us to develop a consistent explanation of a rather vexing aspect of this experiment. We thank Professor Darrell G. Schlom of The Pennsylvania State University and Professor Robert L. Rasera of University of Maryland Baltimore County, who contributed much knowledge and expertise. We thank Professor Tilman Butz of Universität Leipzig for his comments and careful review of the manuscript. We thank the Office of Naval Research for partial support via Grant No. N0014-90-J-4112.